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Simulation and Tomography of Closed-cell Polymer Foam in Compression

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Polymer Mechanics Laboratory

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Outline

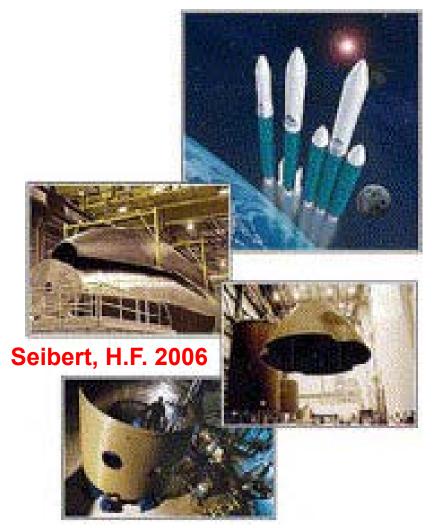
- In-situ micro-computed tomography (µ-CT) and radiography of Rohacell (Polymethacrylimide, PMI) foam under compression
- Mechanical characterization using compression, and nanoindentation
- Simulation using the Material Point Method (MPM)

Applications for PMI Foam

• Rohacell A foam is primarily used in aircraft as cores for composite sandwich structures up to 266 °F and 45 psi

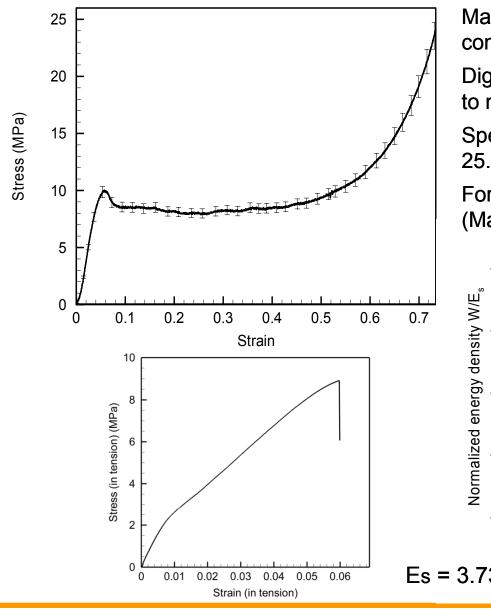
• As cores for sandwich structure applications realized by all composite manufacturing processes in aerospace/aircraft marine, sporting goods, wind energy, medical beds, automotive, electronics, energy absorption for crash protection, and others.

•Sandwich structures where electric conductivity is required within the core structure.



PMI cored components for Delta 4: payload fairing; payload adapter; interstage; centre body; thermal shield; booster nose cones.

Compressive Stress-Strain Curve of PMI Foam

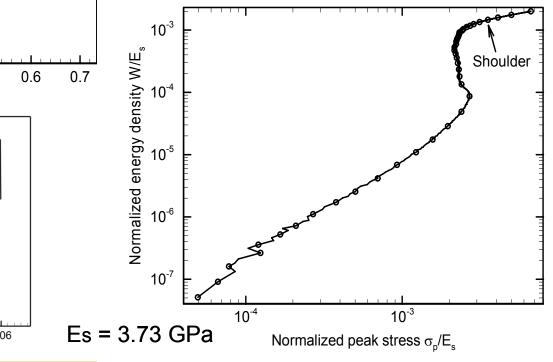


Machine compliance was corrected in compression.

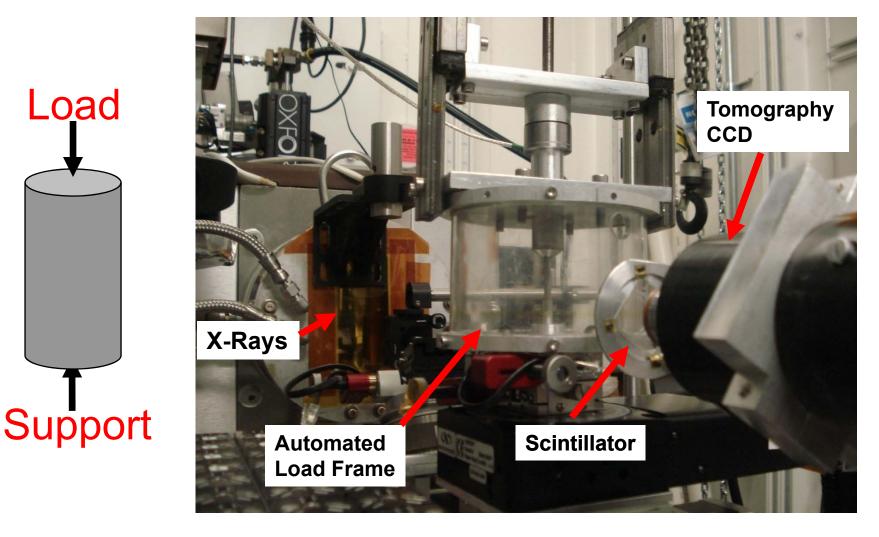
Digital Image Correlation (DIC) was used to measure surface deformations in tension.

Specific energy absorption up to 56% strain: 25.5 J/g, up to 73%: 35.0 J/g.

For a typical high strength dual phase (Martensite/ferrite) steel: 12-15 J/g.

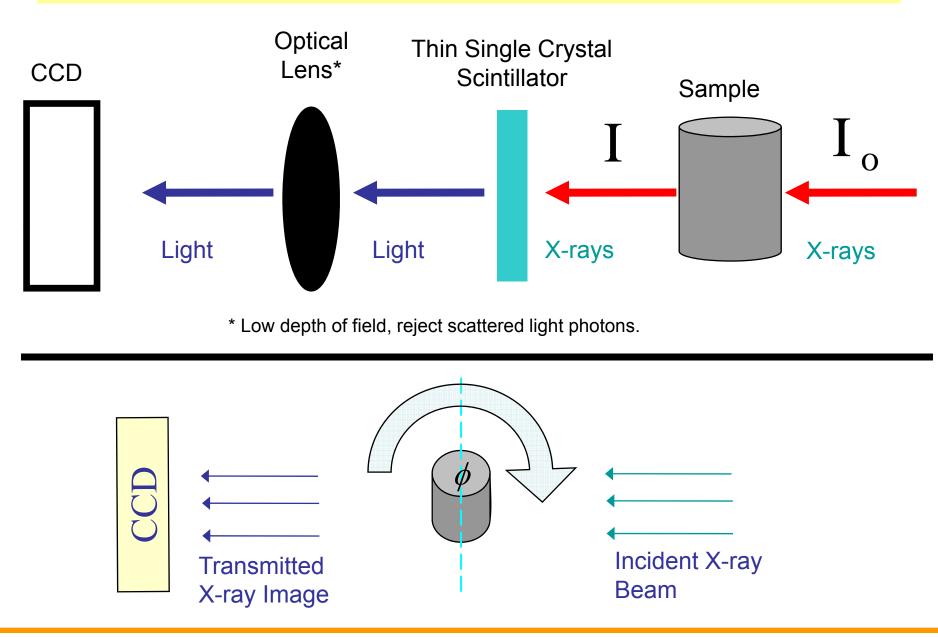


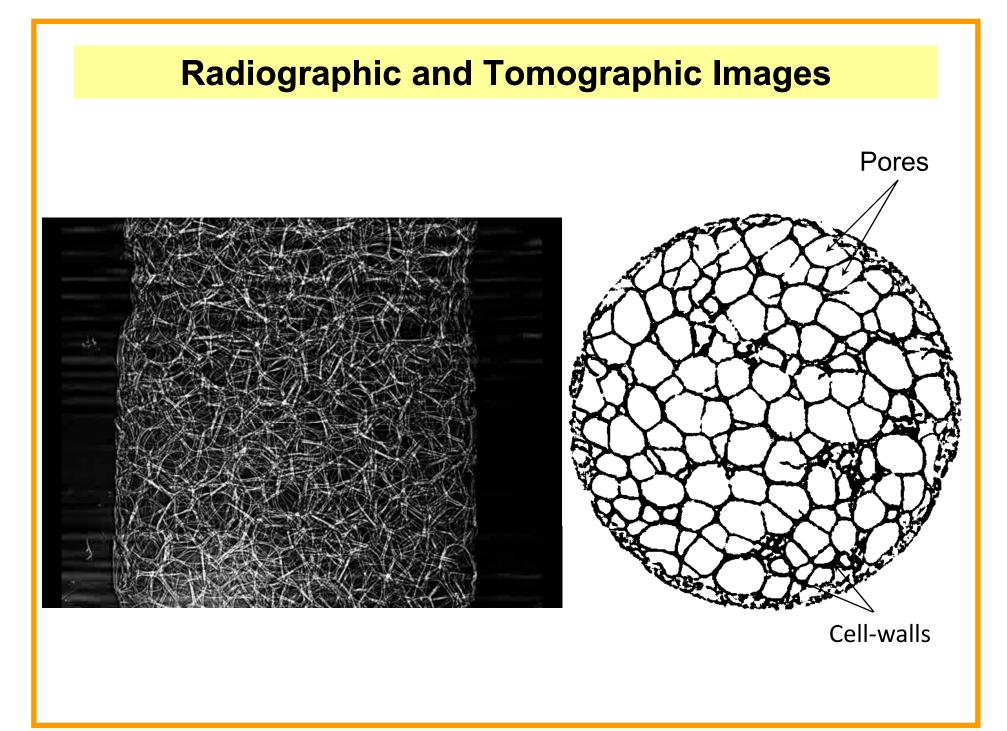
Radiography and Tomography Setup



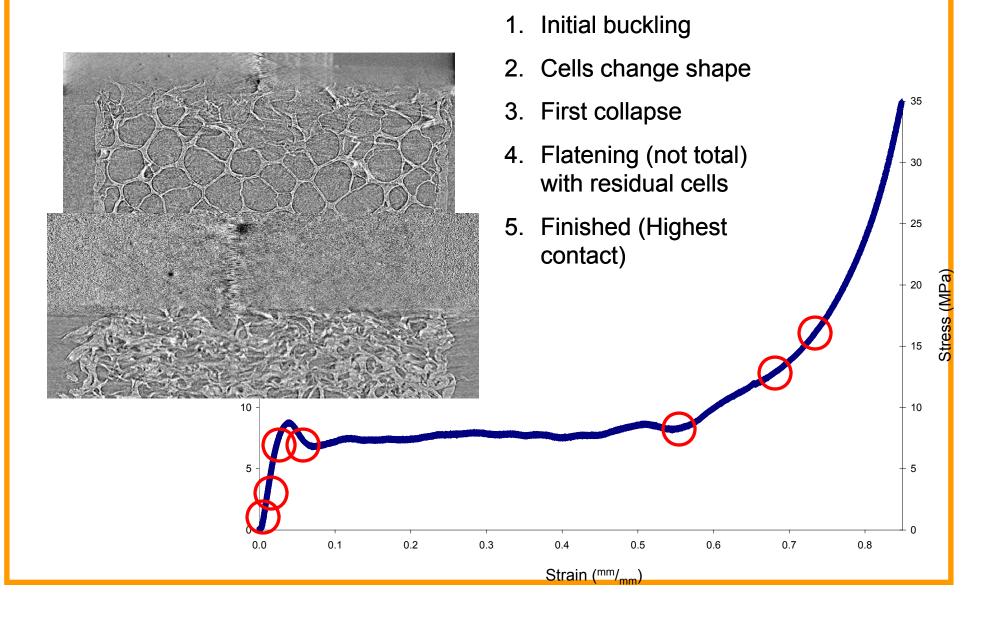
Argonne National Laboratories, 2-BM-B

Micro-Computed Tomography (µ-CT)



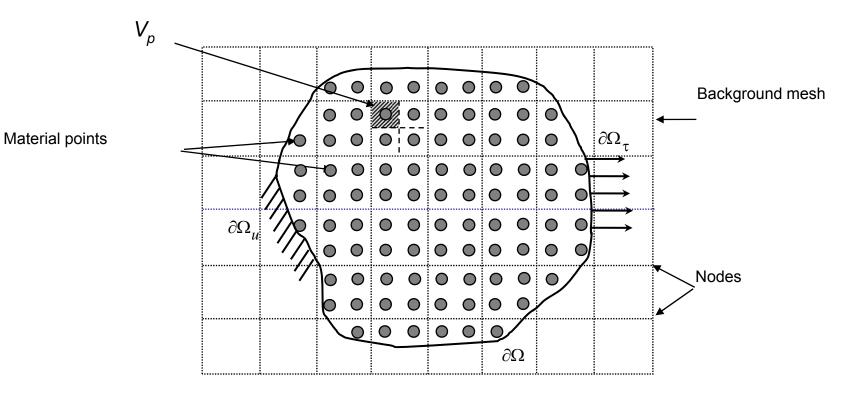


µ-Tomographs at Several Deformed States



Overview of MPM & GIMP

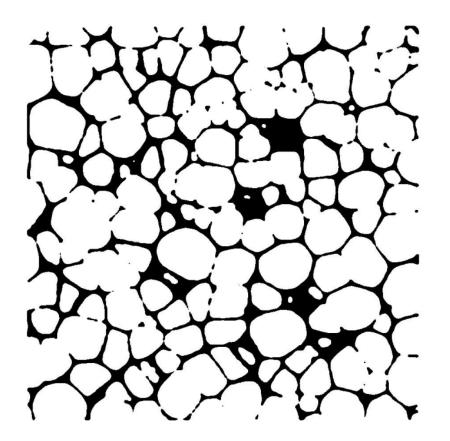
MPM: Material Point Method GIMP: Generalized Interpolation Material Point Method



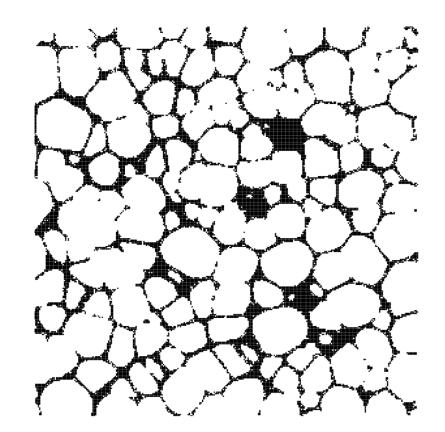
Harlow, 1964; Brackbill *et al.* 1987; Sulsky *et al.* 1995; Bardenhagen and Kober 2000, 2004; Tan and Nairn 2002; York *et al.* 1999; Banerjee, 2005, Bardenhagen *et al.* 2000; Bardenhagen and Brydon, 2005; Bardenhagen and Brackbill, 1998; Ayton *et al.* 2001; Chen and Brannon, 2002; Nairn, 2004; Hu and Chen, 2003; Guilkey *et al.* 2005; Shen and Chen, 2005, Lu *et al.* 2006; Schreyer *et al.* 2006

GIMP Simulation: Reconstruction and Modeling

Discretization: Voxels → Material points



A Section of a Tomographic Image (Grayscale Image)



A Section in the MPM Model (binary)

GIMP Discretized Model

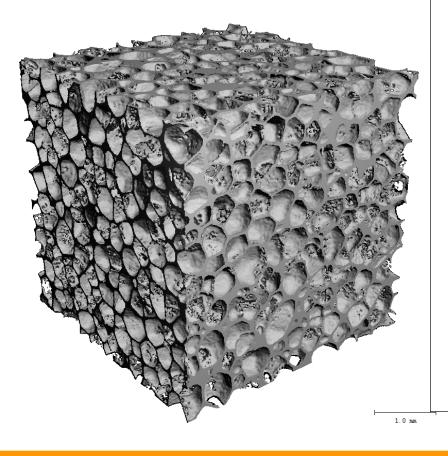
Porosity: 70.6% Avg. Cell size: 0.32 mm Avg. Wall thickness: 0.05 mm Resolution: 8 µm / voxel

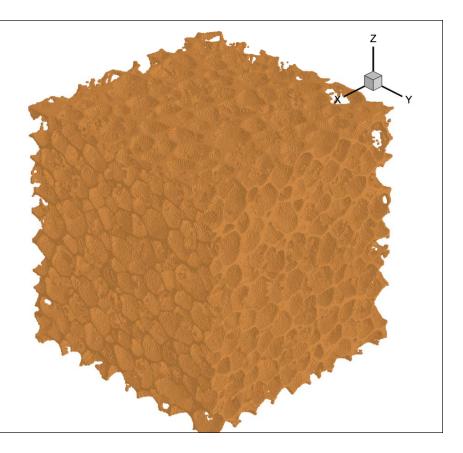
Tomographic Image

Tomographic image acquired

using Scanco μ CT-40 at OSU.

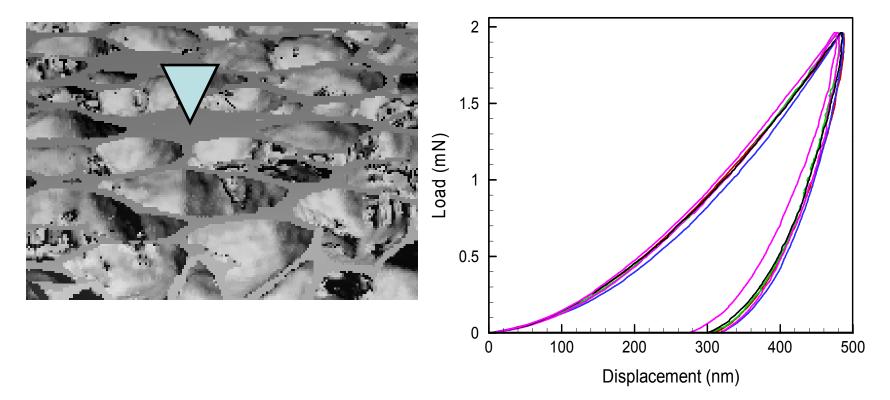
MPM Model





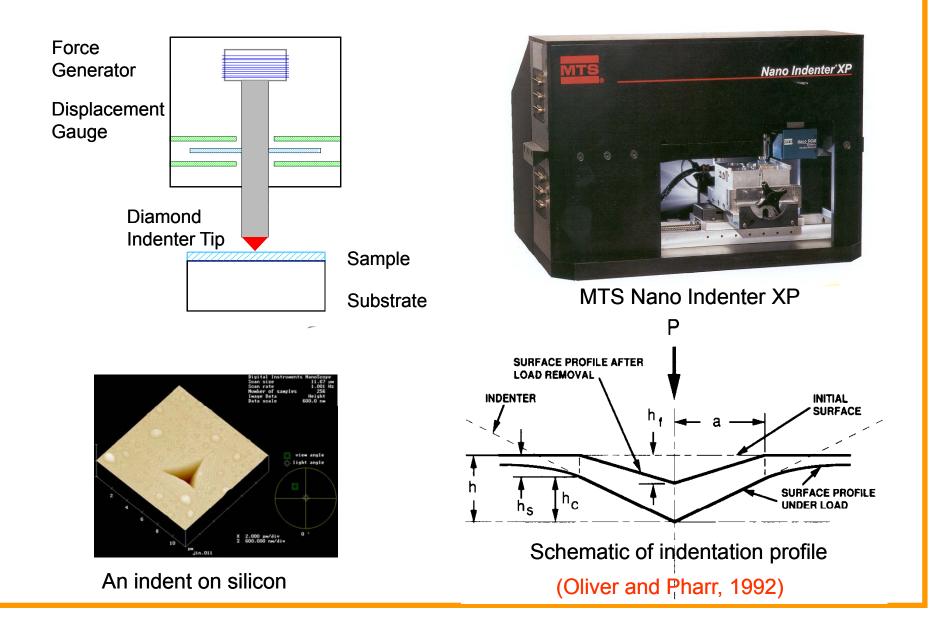
Nanoindentation on the Cell-walls

Direct Measurement of the Mechanical Properties of Cell-walls

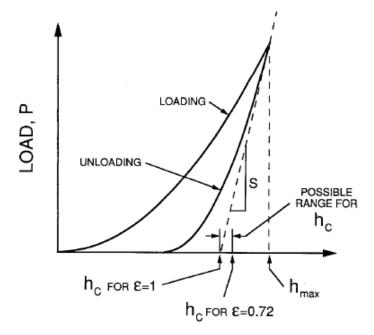


Small pieces of PMI foam were embedded in epoxy. The surface was polished using a minimum abrasive size of 50 nm (Buehler). The sample was annealed to relieve the stresses. A diamond Berkovich indenter tip was used. A constantrate loading history was used in all nanoindentation tests.

Nanoindentation

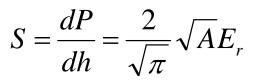


The Oliver-Pharr Approach (1992)



$$P = C(h - h_f)^m$$

[From Sneddon's solution (1965)]



$$\frac{1}{E_r} = \frac{1 - v^2}{E} + \frac{1 - v_i^2}{E_i}$$

A typical nanoindentation P-h curve

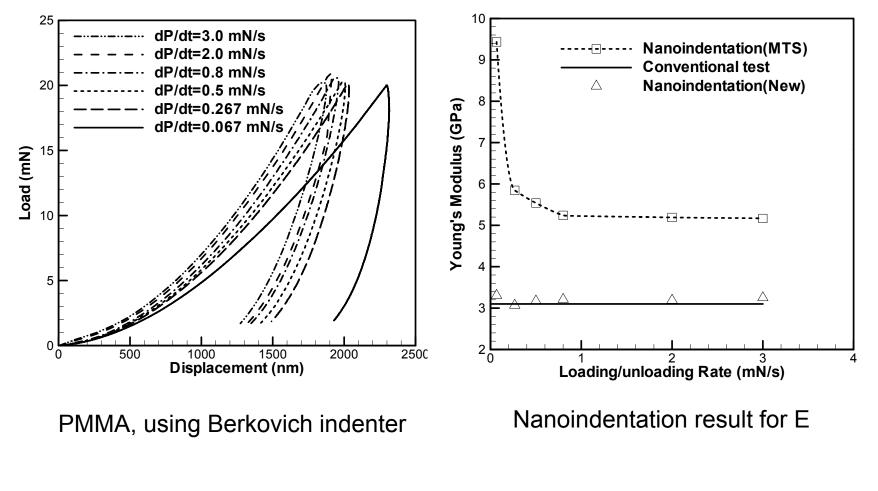
$$A = a_0 h_c^{2} + a_1 h_c + a_2 h_c^{1/2} + a_3 h_c^{1/4} + a_4 h_c^{1/8}$$

Using this approach, the Young's modulus of PMI cell wall was determined as 9.6 GPa, much higher than the actual value, in the neighborhood of 4 GPa.

(Indenter tip calibration)

$$h_c = h - h_s = h - \varepsilon \frac{P}{S}$$

A Viscoelastic Method to Determine the Viscoelastic Properties by Nanoindentation



(Lu, et al., MTDM, 2003)

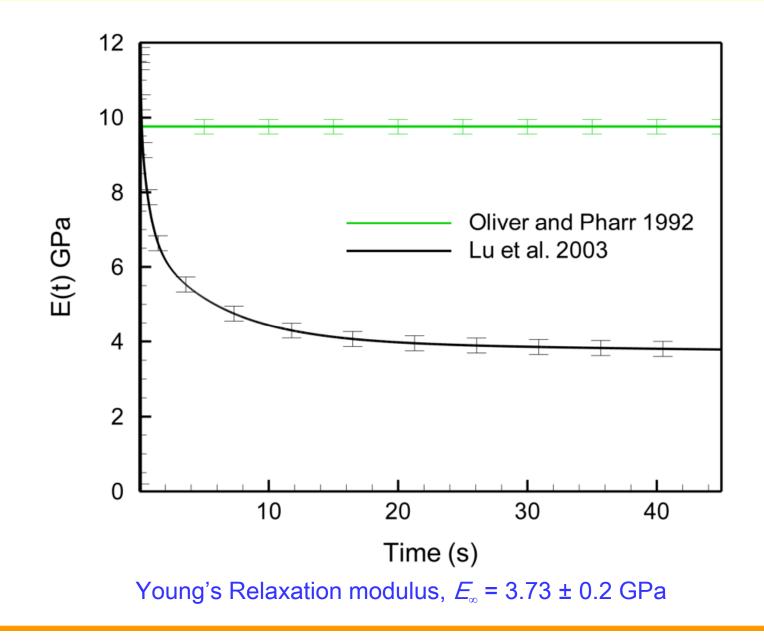
Two Approaches in Nanoindentation to Measure the Creep Compliance

Constant Rate
$$P(t) = v_0 t H(t)$$

Nethod I: $J(t) = \frac{8h}{\pi(1-\nu)\tan\alpha} \frac{dh}{dP}$
Method II: $J(t) = \frac{1}{4}\pi(1-\nu)v_0 \tan\alpha[(J_0 + \sum_{i=1}^n J_i)t - \sum_{i=1}^n J_i\tau_i(1-e^{-\frac{t}{\tau_i}})]$
 $J(t) = J_0 + \sum_{i=1}^n J_i(1-e^{-t/\tau_i})$
Step Loading: $P(t) = P_0 H(t)$
 $J(t) = \frac{4h^2(t)}{\pi(1-\nu)P_0 \tan\alpha}$
 $J(t) = \frac{1}{4}\pi(1-\nu)v_0 \tan\alpha[(J_0 + \sum_{i=1}^n J_i\tau_i(1-e^{-\frac{t}{\tau_i}})]$
 $J(t) = J_0 + \sum_{i=1}^n J_i(1-e^{-t/\tau_i})$
Step Loading in Reality)

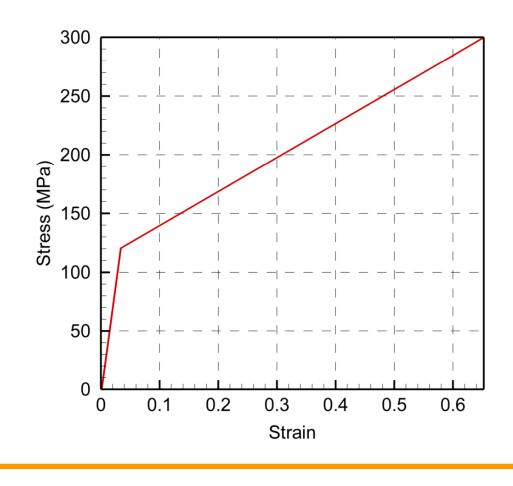
(Berkovich indenter, Poisson's ratio is assumed a constant)

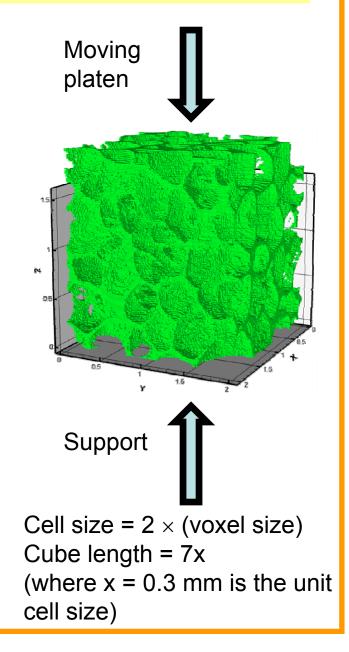
Viscoelastic Properties of Foam Parent Material



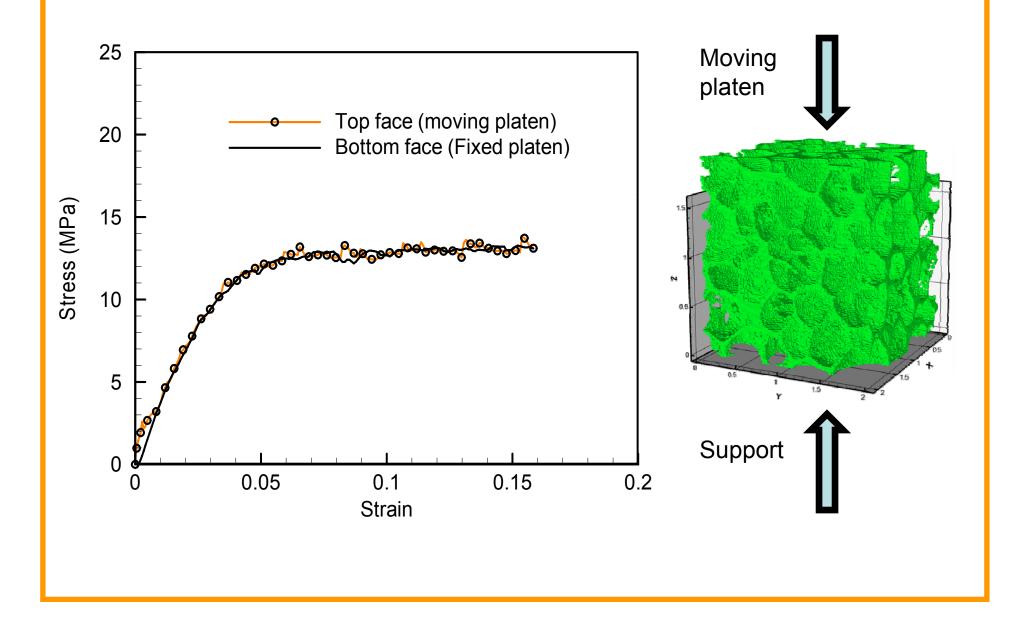
Modeling of PMI Foam under Compression

<u>Constitutive model for parent material</u>: von Mises elastic-plasticity with isotropic bilinear hardening Density: 1.2 g/cc, Poisson's ratio: 0.35

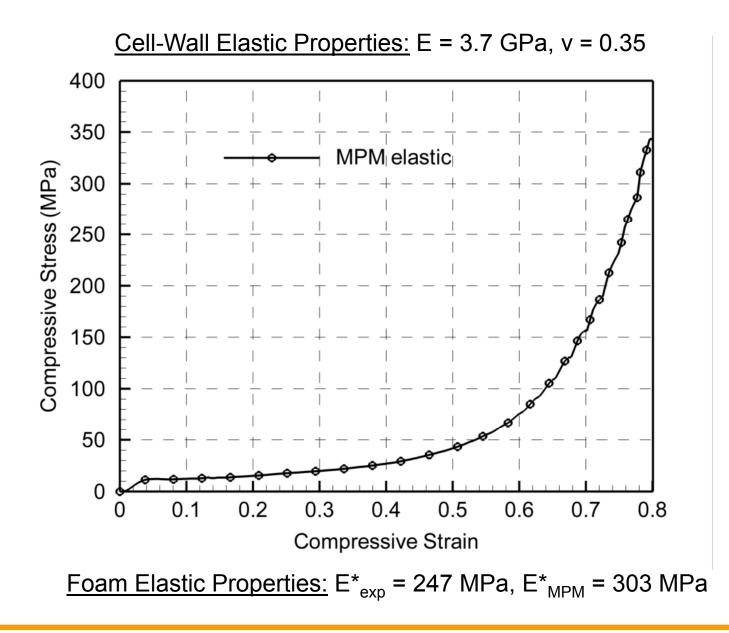




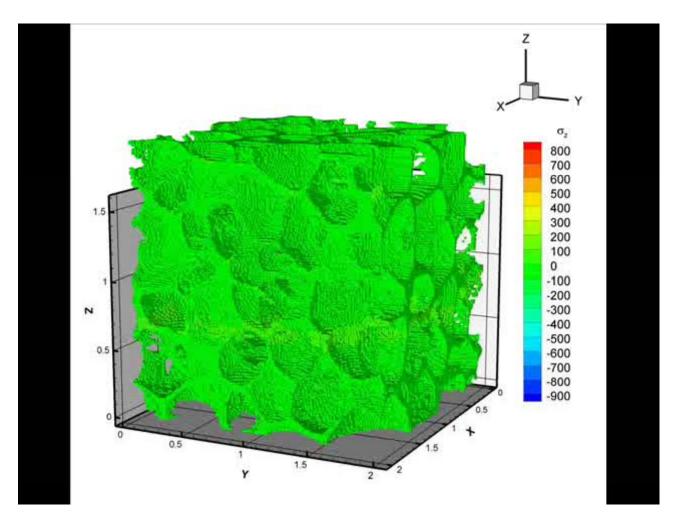
Dynamic Stress Equilibrium Condition



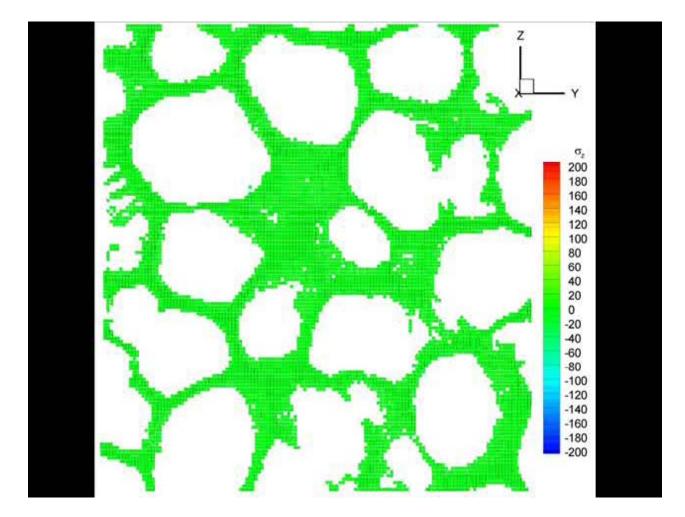
Stress-Strain Curve from GIMP Simulation



Simulation of the Compaction Process (Bi-linear Model)

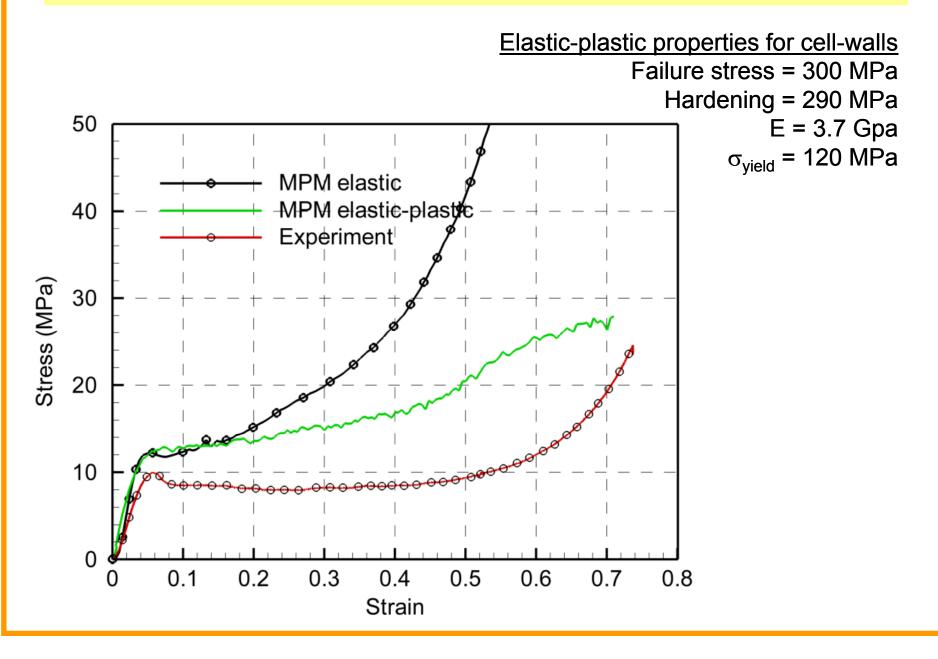


Simulation of the Compaction Process

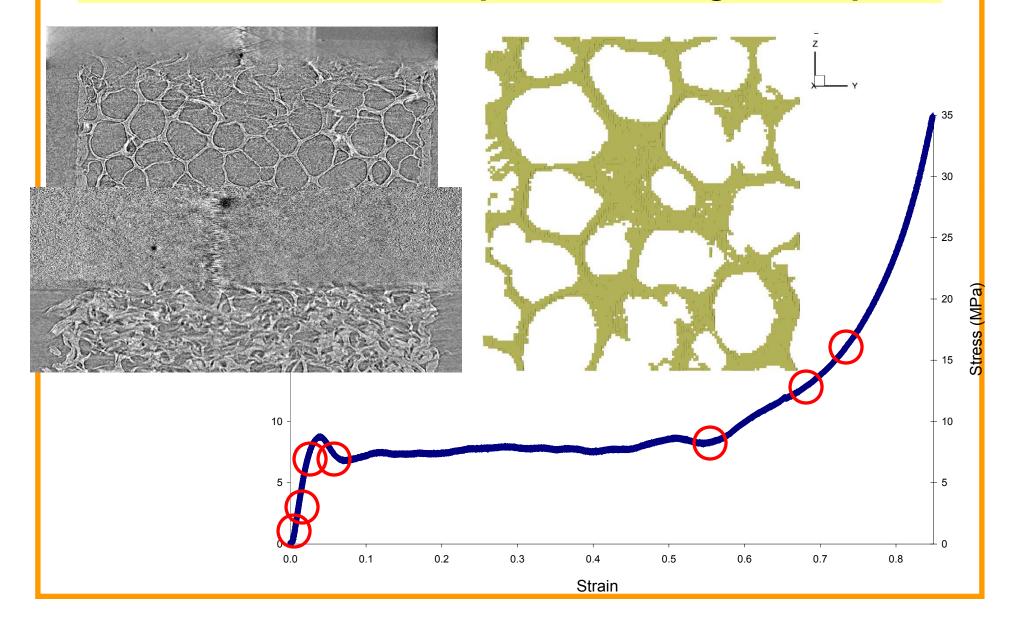


Central section showing stress σ_{zz} (MPa)

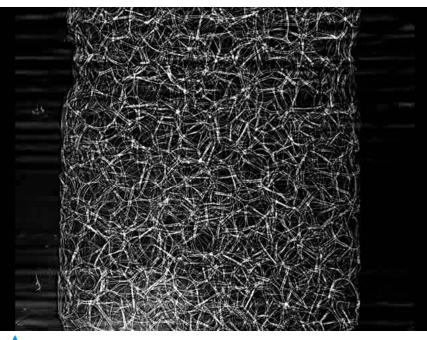
Stress-Strain Curve from GIMP Simulation



Qualitative Comparison of deformation pattern from simulation with experiment using *in-situ* µ-CT

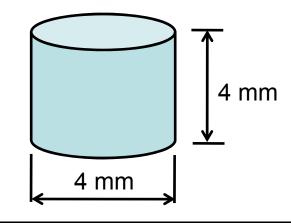


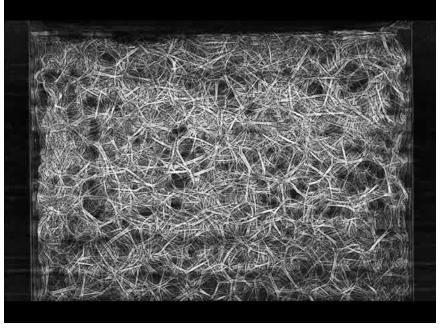
What's Next?



Unconfined compression Quality not high enough for MPM model generation.

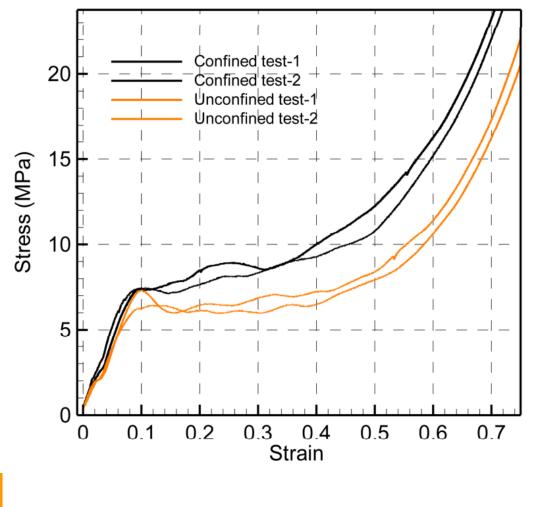
Confined compression -> Preliminary analysis shows that image quality is high.

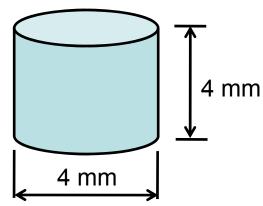




Field of view has 2.8 mm height, a portion of the sample

Other Considerations





RVE is appropriate for homogeneous Behavior, such as elastic modulus.

For failure analysis, such as compaction, RVE is not necessarily representative. We will simulate the entire sample to investigate the failure behavior.

Issues that need to be addressed: The complete stress-strain curve; Nonlinear viscoelasticity; Contact algorithm; Interpolation induced cell-wall thickening; etc.

Concluding Remarks

- In-situ µ-CT tomographic images were acquired during the compression of PMI foam at different strain levels. Tomographic image acquired from Scanco µCT-40 tomography was used to generate MPM model in simulation.
- Viscoelastic nanoindentation was used to determine the steadystate modulus of the cell-walls. The foam elastic modulus was determined with reasonably good agreement. The collapse stress is determined by the elastic buckling of the cell walls, independent of the plastic behavior for the few situations considered.
- The simulated deformed states can capture features in in-situ tomographic images.
- In an attempt to solve an inverse problem, the yield stress and hardening modulus were adjusted in a bi-linear constitutive model, with an attempt to allow the simulated stress-strain curve to have an agreement with experimental data. The drawback of the inverseproblem solving approach is the computational cost.
- Future work is planned to simulate the entire sample under compression, using nonlinear constitutive law determined from nanoindentation measurements.